



A Review of Bifacial Perovskite Solar Cells

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ABSTRACT

Bifacial perovskite solar cells (B-PSCs) have emerged as a promising solution to enhance photovoltaic efficiency by harvesting light from both front and rear surfaces. While perovskite materials offer high power conversion efficiency and low-cost fabrication, traditional single-sided architectures limit their energy-harvesting potential. The purpose of this research is to systematically analyze recent progress in materials design, device architecture, and optical management strategies for B-PSCs. Key results indicate that optimized electrodes and interlayers significantly improve rear-side transparency, responsivity and overall energy yield. B-PSCs represent a viable pathway toward next-generation, high-efficiency photovoltaics, though further work is needed to address stability, large-scale production, and long-term operational performance.

1. Introduction

The growing demand for renewable energy solutions has catalyzed significant advancements in photovoltaic technologies, with perovskite solar cells (PSCs) emerging as one of the most promising alternatives to traditional silicon-based systems due to their remarkable power conversion efficiencies (PCEs), low-cost fabrication processes, and material versatility [1]. Among the various device architecture in this field, bifacial perovskite solar cells (B-PSCs) have gained considerable attention due to their ability to harvest light from both front and rear surfaces, effectively enhancing total energy generation [2].

The core theoretical advantage of B-PSCs lies in their ability to exploit direct, diffuse, and ground-reflected light, phenomena explained by radiative transfer theory and the albedo effect, as illustrated in Figure 1. The latter refers to the fraction of solar radiation reflected from the ground or nearby surfaces and captured through the rear side of the solar cell. This dual-side illumination enables higher photocurrent generation compared to traditional monofacial architectures, thus expanding their operational efficiency in real-world environments such as building-integrated photovoltaics (BIPVs) [3], floating solar systems [4], and agrivoltaics installations [5].

In addition to outdoor applications, recent research has underscored the remarkable performance of B-PSCs under indoor lighting conditions. The high absorption coefficient, tunable

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bandgap, and low-light response of perovskite materials make them highly suitable for energy harvesting in artificial or low-intensity environments. This opens new possibilities for B-PSCs in powering indoor Internet-of-Things (IoT) devices and self-powered electronics [6,7]. Notable studies have demonstrated promising PCE under indoor LED illumination, reinforcing the versatility of B-PSCs for multifunctional energy solutions [8,9]

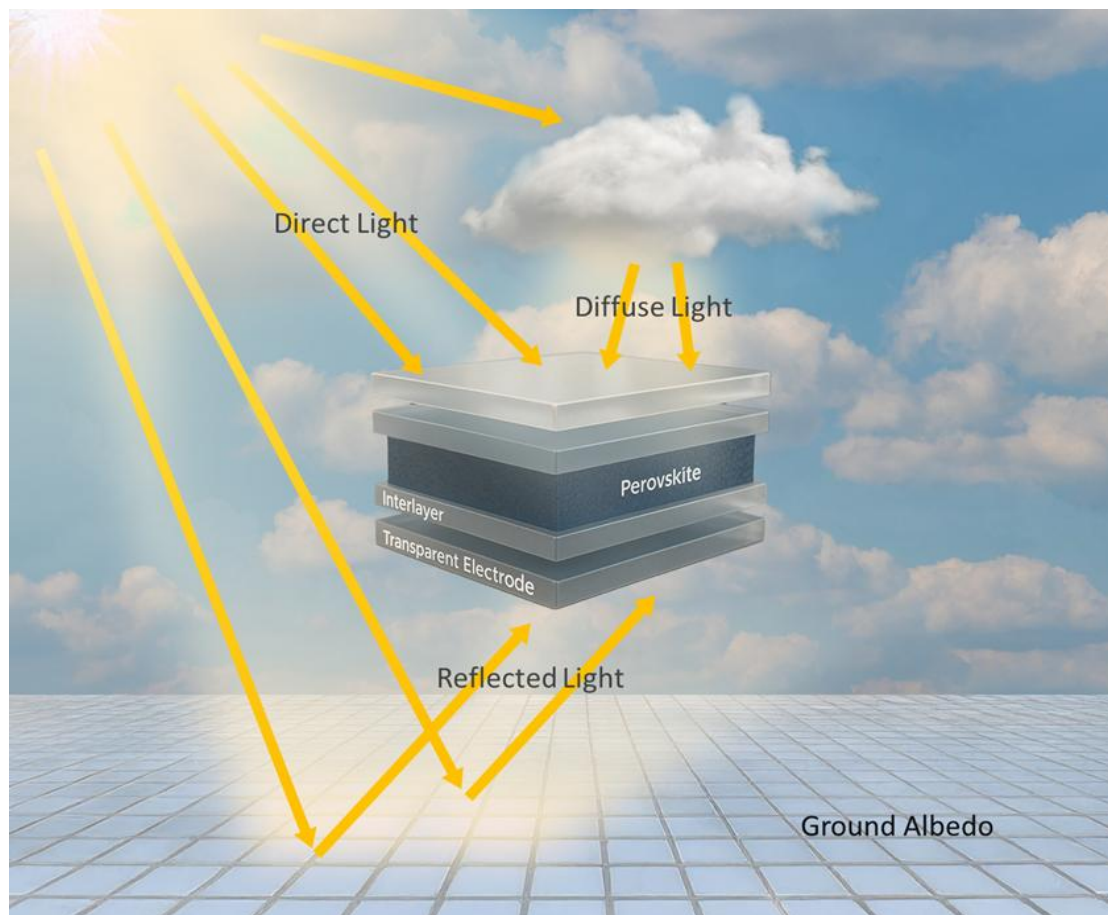


Fig. 1. Schematic illustration of a bifacial perovskite solar cell receiving light from both the front and rear sides, including direct sunlight, diffuse sky radiation, and ground-reflected light due to the albedo effect

B-PSCs exploit the unique optoelectronic properties of perovskite materials, which contribute to their high efficiency and versatility. One of the most notable features is their high absorption coefficient, which allows them to capture a larger portion of the solar spectrum, enhancing their PCE. In addition, B-PSCs demonstrate exceptional defect tolerance, meaning they are less affected by imperfections in the material, ensuring stable performance over time. They also possess long carrier lifetimes and extended carrier diffusion lengths, which enable charge carriers to travel further without recombining, ultimately improving efficiency. The low surface recombination in B-PSCs minimizes energy losses at the interfaces, while passivated interfaces help reduce defects and increase the overall device stability. Furthermore, the tunable band alignment of B-PSCs allows for optimization of the energy levels, making them adaptable to a range of light conditions and improving their overall performance.

B-PSCs also demand advanced structural engineering to achieve effective charge separation under dual-illumination. This typically involves a symmetrical or semi-transparent stack comprising transparent conductive oxides, passivated charge transport layers, and optimized band alignment to

facilitate charge collection from both surfaces [10,11]. Moreover, B-PSCs are made from abundant and cost-effective materials, and their improved efficiency reduces the need for excessive material use, resulting in lower production and operational costs. This combination of higher energy output, increased efficiency, durability, and economic advantages positions B-PSCs as a compelling alternative to traditional monofacial cells in the rapidly expanding solar energy market.

This review provides a comprehensive overview of B-PSCs, exploring their structure, performance, and the underlying mechanisms that influence their efficiency. Despite these promising attributes, the development of B-PSCs faces several challenges, including issues related to material stability, carrier recombination, and the overall efficiency. By delving into the latest research and technological advancements, we aim to highlight the potential of B-PSCs as a transformative technology in the pursuit of more efficient and cost-effective solar energy solutions.

2. Working Principle of B-PSCs

2.1 Light Absorption, Charge Generation and Transport

B-PSCs operate by harnessing incident light from both the front and rear surfaces of the device, thereby extending their photoconversion potential beyond that of conventional monofacial designs. The core of the device comprises a organic-inorganic hybrid perovskite such as methylammonium lead iodide, MAPbI_3 ($\text{CH}_3\text{NH}_3\text{PbI}_3$) and formamidinium lead iodide, FAPbI_3 ($\text{CH}(\text{NH}_2)_2\text{PbI}_3$), or all-inorganic perovskite such as caesium lead iodide (CsPbI_3) or its multi-cation variants which serves as the light-absorbing material due to its direct bandgap (typically $\sim 1.5\text{--}1.8$ eV) and exceptionally high absorption coefficients ($>10^5 \text{ cm}^{-1}$) [12-15].

Upon photon absorption, electron-hole pairs are generated and rapidly dissociate into free carriers due to the low exciton binding energy ($\sim 10\text{--}50$ meV) intrinsic to perovskite materials. The efficient separation and collection of these carriers rely on a carefully engineered heterojunction stack. Electrons are typically transported through an electron transport layer (ETL) such as titanium dioxide (TiO_2), tin dioxide (SnO_2), or zinc oxide (ZnO) while holes migrate through a hole transport layer (HTL), often composed of Spiro-OMeTAD, Poly[bis(4-phenyl)(2,4,6-trimethylphenyl)amine] (PTAA), or nickel oxide (NiO_x) [13, 16-18]. The energy level alignment between the perovskite and transport layers is critical for minimizing potential barriers and recombination losses.

In bifacial architectures, both electrodes must be at least partially transparent to facilitate dual-side illumination. Transparent conductive oxides (TCOs) such as indium tin oxide (ITO) or fluorine-doped tin oxide (FTO) are typically employed at one or both interfaces. Charge collection is further influenced by the diffusion length and carrier mobility in the perovskite film, as well as the conductivity of the electrode. Additionally, bifacial designs often incorporate optical enhancement strategies such as photonic structures to maximize light trapping and internal photon recycling, thus enhancing charge generation without compromising extraction efficiency.

2.2 Performance Metrics in B-PSCs

The evaluation of B-PSCs extends beyond the conventional efficiency metrics used for monofacial solar cells due to their unique dual-illumination capability. While PCE remains a foundational metric, a more comprehensive performance characterization must account for additional parameters that reflect dual-sided light harvesting. These include the bifacial factor (BF), which quantifies the ratio of rear-side to front-side efficiency; the bifacial gain (BG), which measures the net improvement in power output compared to monofacial devices; and the energy yield (kWh/kWp/day), which captures long-term power generation under real-world conditions.

2.2.1 Power conversion efficiency (PCE)

PCE is calculated using the standard formula following Eq. (1):

$$PCE = \frac{J_{sc} \times V_{oc} \times FF}{P_{in}} \quad (1)$$

where, J_{sc} is short-circuit current density, V_{oc} is open-circuit voltage, FF is fill factor and P_{in} is the incident power density typically 100 mW/cm² if measured under AM1.5G 1 sun illumination. In B-PSCs, this value is typically determined separately for the front and rear surfaces due to varying optical paths and incident angles.

2.2.2 Bifacial factor (BF)

To capture the performance symmetry of bifacial devices, the BF is defined as shown in Eq. (2):

$$BF (\%) = \frac{PCE_{rear}}{PCE_{front}} \times 100 \quad (2)$$

BF provides a dimensionless indication of rear-side utilization efficiency. High BF values (typically above 85%) are indicative of well-optimized rear electrode designs with minimal optical and electrical losses.

2.2.3 Bifacial gain (BG)

BG quantifies the net improvement in power output when both sides are utilized compared to a standard monofacial cell. BG is expressed as the percentage increase in energy yield attributable to rear-side illumination under standard testing conditions. This metric is particularly important for real-world evaluations where diffuse and ground-reflected light (characterized through the albedo effect) can significantly enhance overall energy production as shown by Eq. (3):

$$BG (\%) = \frac{P_{bifacial} - P_{monofacial}}{P_{monofacial}} \times 100 \quad (3)$$

where $P_{bifacial}$ is the power output or energy yield from the bifacial solar cell (front and rear) and $P_{monofacial}$ is the power output or energy yield from a monofacial solar cell (front only).

2.2.4 Energy yield

Energy yield typically expressed in kWh/kWp/day or kWh/kWp/year serves as a pivotal performance indicator for B-PSCs. This metric encapsulates the cumulative electrical output over extended temporal scales, inherently accounting for diurnal and seasonal variations in solar irradiance, temperature, and environmental conditions. Notably, energy yield reflects the tangible benefits of dual-sided light collection, especially in deployment scenarios where ground-reflected irradiance (albedo) significantly enhances rear-side illumination. As such, elevated energy yields in B-PSCs substantiate their potential for high-efficiency deployment in real-world applications.

3. Performance of B-PSCs

Over the past decade, B-PSCs have demonstrated rapid progress in efficiency and architecture refinement. Table 1 shows the recent progress in B-PSCs technology for 10 years starting from 2015 to 2025.

Table 1

Summary of bifacial PSCs with PCE from front and rear side with bifacial factor

Author/ Year	Device Structure	PCE (%)		BF (%)
		Front	Rear	
Fu <i>et al.</i> , [19]	FTO/ZnO/PCBM/CH ₃ NH ₃ PbI ₃ /MoO ₃ /In ₂ O ₃ :H	14.2	9.6	67.60
Xiao <i>et al.</i> , [20]	ITO/bl-TiO ₂ /m-TiO ₂ / CH ₃ NH ₃ PbI ₃ /FTO	8.67	8.27	95.38
	ITO/bl-TiO ₂ /m-TiO ₂ / CH ₃ NH ₃ PbI ₃ /PEDOT/FTO	12.33	11.88	96.35
Pang <i>et al.</i> , [21]	ITO/PEDOT:PSS/ CH ₃ NH ₃ PbI ₃ /PCBM/Ag	8.04	5.24	65.11
	ITO/PEDOT:PSS/ CH ₃ NH ₃ PbI ₃ /PCBM/Ag/MoO _x	10.40	6.54	62.88
	ITO/PEDOT:PSS/ CH ₃ NH ₃ PbI ₃ /PCBM/PEIE/Ag/MoO _x	13.55	8.41	62.06
	ITO/PEDOT:PSS/CH ₃ NH ₃ PbI ₃ /PCBM/PEIE/Ag/MoO _x /back reflector	14.5	11.37	78.41
Hanmandlu <i>et al.</i> , [22]	ITO/PEDOT:PSS/ CH ₃ NH ₃ PbI ₃ /BCP/Ag/MoO ₃	13.49	9.61	71.23
Fan <i>et al.</i> , [23]	FTO/TiO ₂ /FA _{0.5} MA _{0.5} PbI _{3-x} Cl _x /CuSCN/Au	12.47	8.74	70.09
Martinez-D. <i>et al.</i> , [24]	ITO/SnO ₂ /TiO ₂ / CH ₃ NH ₃ PbI ₃ /PTAA/Au	12.9	9.1	70.54
Deluca <i>et al.</i> , [25]	FTO/TiO ₂ /Cs _{0.05} FA _{0.79} MA _{0.16} PbI _{2.49} Br _{0.51} /CuSCN/ITO	1.7	1.4	82.35
Pang <i>et al.</i> , [26]	ITO/PEDOT:PSS/Cs _{0.05} FA _{0.3} MA _{0.7} PbI _{2.51} Br _{0.54} /PCBM/BCP/Ag/V ₂ O ₅	14.01	8.91	63.60
	ITO/PEDOT:PSS/ Cs _{0.05} FA _{0.3} MA _{0.7} PbI _{2.51} Br _{0.54} /PCBM/BCP/Ag/V ₂ O ₅ /reflector	15.39	12.44	80.83
Li <i>et al.</i> , [27]	FTO/c-TiO ₂ /m-TiO ₂ /CsPbBr ₃ /carbon/CsPbBr ₃ /m-TiO ₂ /c-TiO ₂ /FTO	7.55	7.44	98.54
Lee <i>et al.</i> , [28]	FTO/c-TiO ₂ /m-TiO ₂ / CH ₃ NH ₃ PbI ₃ /CT/MoO _x /ITO	14.96	13.61	90.98
Chiang <i>et al.</i> , [29]	FTO/TiO _x /mp TiO ₂ / CH ₃ NH ₃ PbI ₃ /Spiro/MoO _x /IZO	16.40	15.26	93.04
Chen <i>et al.</i> , [30]	ITO/NiO _x /FA _{0.3} MA _{0.7} PbI _{3-x} Cl _x /PC ₆₁ BM/BCP/Ag	12.7	7.33	57.72
	ITO/NiO _x /FA _{0.3} MA _{0.7} PbI _{3-x} Cl _x /PC ₆₁ BM/BCP/Ag/TeO ₂	18.86	15.12	80.17
	ITO/NiO _x /FA _{0.3} MA _{0.7} PbI _{3-x} Cl _x -PEAI/ PC ₆₁ BM / BCP/Ag	16.07	8.40	52.27
	ITO/NiO _x /FA _{0.3} MA _{0.7} PbI _{3-x} Cl _x -PEAI/ PC ₆₁ BM / BCP/TeO ₂ /Ag	20.25	16.75	82.72
	ITO/NiO _x /FA _{0.3} MA _{0.7} PbI _{3-x} Cl _x -PEAI/ PC ₆₁ BM / BCP/Ag/TeO ₂ -with glass reflector	22.51	21.49	95.47
	ITO/NiO _x /FA _{0.3} MA _{0.7} PbI _{3-x} Cl _x -PEAI-PEAI/ PC ₆₁ BM / BCP/Ag/TeO ₂	12.42	10.55	84.94
	ITO/SnO ₂ /CH ₃ NH ₃ PbI ₃ /NiO _x /ITO	8.47	7.10	83.83
Li <i>et al.</i> , [32]	ITO/SnO ₂ /CH ₃ NH ₃ PbI ₃ /undoped Spiro-OMeTAD/sorbitol/PEDOT:PSS/ITO	15.02	13.79	91.81
Liang <i>et al.</i> , [33]	ITO/np-SnO ₂ / CH ₃ NH ₃ PbI ₃ /Spiro-OMeTAD/MoO _x /Ag/WO ₃	15.40	9.70	62.99
Chen <i>et al.</i> , [34]	FTO/TiO ₂ /CsPbI ₂ Br ₂ /Spiro-OMeTAD/Ag/TeO ₂	8.46	6.40	75.65
Huo <i>et al.</i> , [35]	ITO/SnO ₂ /CsMAFA/CuSCN/MoO _x /ITO/Au	14.80	12.50	84.46
Truong <i>et al.</i> , [36]	ITO/SnO ₂ /Cs _{0.05} FA _{0.80} MA _{0.15} PbI _{2.75} Br _{0.25} /HND-NAr ₂ /MoO ₃ /Au/MoO ₃	10.90	6.80	62.38
	ITO/SnO ₂ /Cs _{0.05} FA _{0.80} MA _{0.15} PbI _{2.75} Br _{0.25} /	12.40	7.80	62.90

	HND-DTP/MoO ₃ /Au/MoO ₃			
	ITO/SnO ₂ /Cs _{0.05} FA _{0.80} MA _{0.15} PbI _{2.75} Br _{0.25} /	11.60	6.10	52.58
	HND-Cbz/MoO ₃ /Au/MoO ₃			
	ITO/SnO ₂ /Cs _{0.05} FA _{0.80} MA _{0.15} PbI _{2.75} Br _{0.25} /	13.0	7.60	58.46
	Spiro-OMeTAD/MoO ₃ /Au/MoO ₃			
Park <i>et al.</i> , [37]	ITO/GO/CH ₃ NH ₃ PbI ₃ /C ₆₀ /PEIE/Ag	12.48	11.45	91.74
Elakshar <i>et al.</i> , [38]	ITO/SnO ₂ /PCBA/Cs _{0.12} FA _{0.88} PbI ₃ /SWCNT	16.30	6.90	42.33
Heo <i>et al.</i> , [39]	ITO/PTAA/Cs _{0.05} (FA _{0.92} MA _{0.08}) _{0.95} Pb(I _{0.92} Br _{0.08}) ₃ /C ₆₀ /BCP/SnO ₂ /ITO	17.68	16.01	90.55
Jeong <i>et al.</i> , [40]	PI@GR(APSIM)/PEDOT:PSS/ CH ₃ NH ₃ PbI ₃ /CH(NH ₂) ₂ PbI ₃ /C ₆₀ /BCP/PI@GR (TETA)	15.10	13.80	91.39
Najafi <i>et al.</i> , [41]	ITO/NiO np + PTAA/Cs _{0.15} FA _{0.85} Pb(I _{0.92} Br _{0.08}) ₃ /PCBM/ZnO np/ZnO/ITO	17.20	16.70	97.09
	ITO/NiO np + PTAA/ Cs _{0.15} FA _{0.85} Pb(I _{0.92} Br _{0.08}) ₃ /PCBM/Zno np/ZnO/ITO	15.80	14.90	94.30
Zhang <i>et al.</i> , [42]	ITO/SnO ₂ /Cs _{0.05} FA _{0.85} MA _{0.10} Pb(I _{0.97} Br _{0.03}) ₃ /Spiro-OMeTAD/MWCNT/ITO/MgF ₂ /IZO/ZnO/CdS/CuInSe ₂ /Mo/Glass	22.20	10.80	48.65
	ITO/ SnO ₂ /Cs _{0.05} FA _{0.85} MA _{0.10} Pb(I _{0.97} Br _{0.03}) ₃ /Spiro-OMeTAD/SWCNT/ITO//MgF ₂ /IZO/ZnO/CdS/CuInSe ₂ /Mo/Glass	21.40	16.80	78.50
Fan <i>et al.</i> , [43]	ITO/SnO ₂ /CH ₃ NH ₃ PbI ₃ /WS ₂ + Spiro-OMeTAD/Ag	19.87	15.48	77.91
Jiang <i>et al.</i> , [44]	FTO/MeO-2PACZ/Rb _{0.05} Cs _{0.05} MA _{0.05} FA _{0.85} Pb(I _{0.95} Br _{0.05}) ₃ /LiF/C ₆₀ /SnO ₂ /IZO	21.40	20.01	93.50
Zhang <i>et al.</i> , [45]	SWCNT/ Cu:NiOx/Cs _{0.05} FA _{0.80} MA _{0.15} Pb(I _x Br _{1-x}) ₃ /SnO ₂ /PCBM/SWCNT	18.54	18.22	98.27
Han <i>et al.</i> , [46]	FTO/c-TiO ₂ /SnO ₂ -Cl/ (FA _{0.95} Cs _{0.05})PbI ₃) _{0.975} (MAPbBr ₃) _{0.025} /Spiro-OMeTAD/MoO ₃ /ITO/Ag/PMMA/MgF ₂	24.12	21.37	88.60

Albedo factors are one of the important criteria in enhancing the PCE of B-PSCs. The albedo factor was determined as the fraction of incoming solar radiation reflected by each surface [47]. Zhang *et al.*, [45] evaluated the performance of their B-PSCs under various environmental condition with different albedo factors. Figure 2a - 2c shows the angles of light received by B-PSCS that are installed with tilted, horizontal and vertical installing configuration. Figure 2d - 2k shows several common reflection surfaces that contribute to reflected illumination. These surfaces include snow, fiberglass, dry and wet soil, concrete, tile, sand, and grass. Among them, snow exhibited the highest albedo factor with 96% within the wavelength range of 300 to 850 nm, as shown in Figure 3. This followed by fiberglass with 67%, grass 37%, dry soil 34%, sand 28%, wet soil 19%, concrete 18% and tile 12%, respectively.

Structural engineering plays the most crucial role in enhancing the transparency and optical performance of rear electrodes in B-PSCs. Various approaches have been explored, including the use of Fabry–Pérot resonant cavities and Bragg reflectors, which manipulate interference effects to maximize light transmission and reflection [48-55]. Other techniques include the incorporation of photonic crystals and nano-patterned dielectric layers, all of which can be tailored to manage light propagation and minimize parasitic absorption in the rear contact. Pang *et al.*, [21] employed a PEIE/Ag/MoO_x trilayer as the rear electrode in their device architecture. Hanmandlu *et al.*, [22] utilized a BCP/Ag/MoO₃ configuration, while Liang *et al.*, [33] adopted a MoO_x/Ag/MoO_x structure to enhance rear-side transparency and stability.

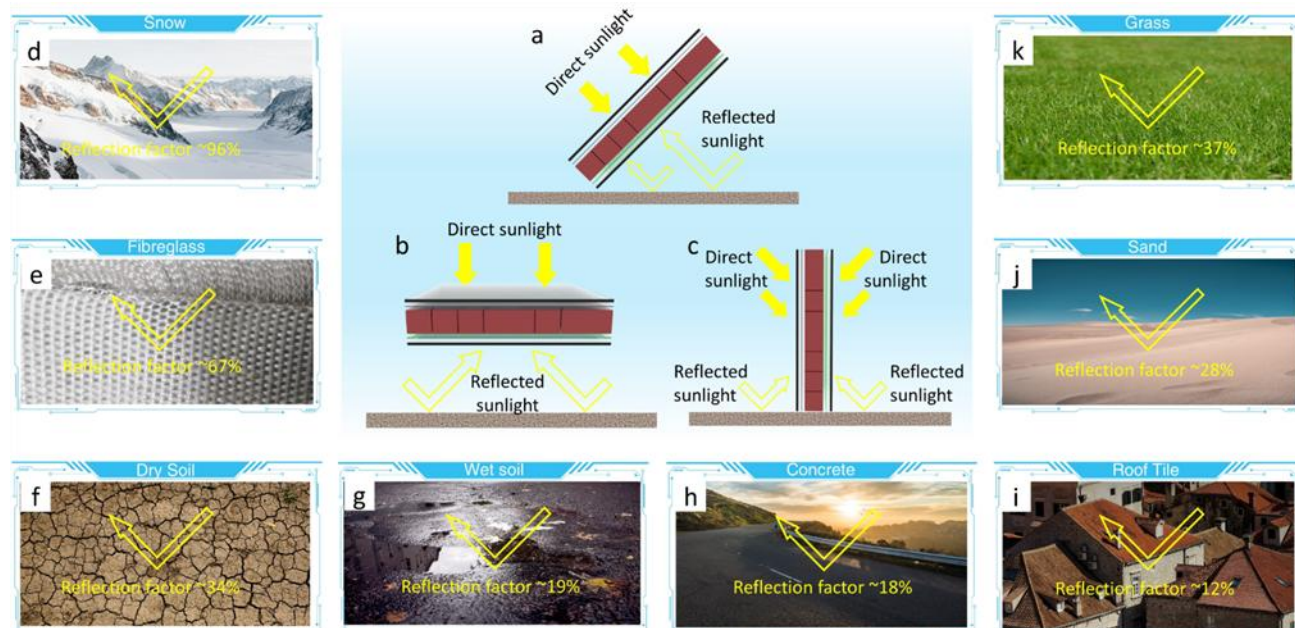


Fig. 2. Angle of light from (a) Titled (b) Horizontal (c) Vertical installing configuration. Reflection surface of (d) Snow (e) Fibreglass (f) Dry soil (g) Wet soil (h) Concrete (i) Roof tile (j) Sand (k) Grass. Credit to photo captured by Erol Ahmed, Mike Erskine, Jens Freudenau, Joshua Sortino, Maxim Berg, Colin Lloyd on Uplash. Reproduced under CC-BY license. Copyright 2024 Springer Nature [45]

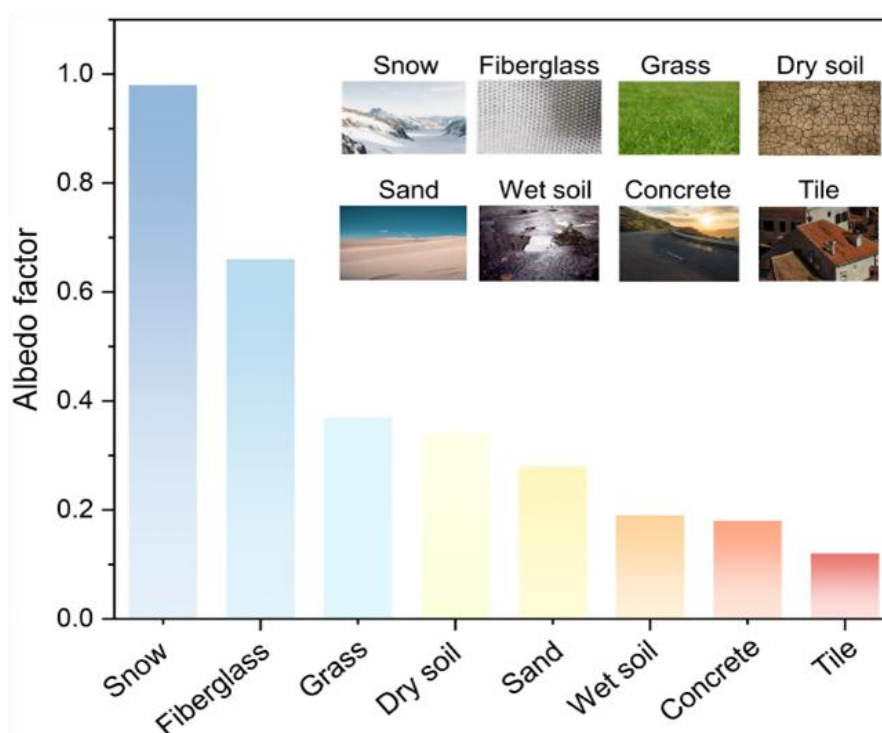


Fig. 3. Albedo factor of eight common ground. Reproduced under CC-BY license. Copyright 2024 Springer Nature [45]

Figure 4 presents the structural and morphological characteristics of a B-PSCs incorporating a BCP/Ag/MoO₃ rear electrode. The schematic in Figure 4a outlines the overall device architecture, highlighting its light penetrate from direct illumination and reflected light from albedo effect. A cross-sectional SEM image (Figure 4b) reveals the layered configuration of the cell, confirming the integration of the BCP/Ag/MoO₃ stack with its film shows highest transparency. Figure 4c and 4d

show SEM surface images of BCP/Ag films with varying BCP/Ag silver thicknesses, demonstrating changes in film uniformity and morphology; inset photographs further illustrate the corresponding optical transparency of each film on glass substrates. Lastly, Figure 4e displays the fabricated semitransparent device, visually confirming the transparency and structural integrity of the electrode stack, which is critical for enabling rear-side light harvesting in bifacial configurations. This configuration resulting PCE of 13.49% from front-side and 9.61% from rear-side resulting BF with 71.23% [22].

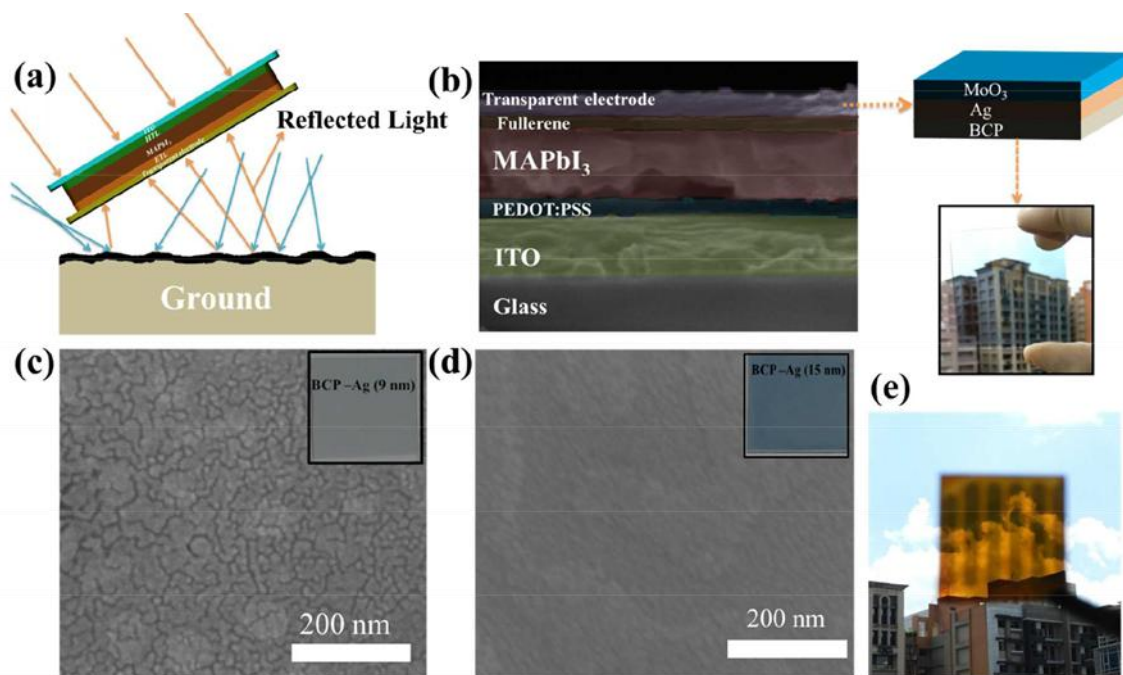


Fig. 4. Structural and morphological characteristics of a B-PSCs incorporating a BCP/Ag/MoO₃ rear electrode (a) Illustration of the device architecture (b) Cross-sectional SEM for whole device and film image of BCP/Ag/MoO₃ layer (c) and (d) SEM surface images of BCP 8 nm/Ag 9 nm and BCP 8 nm/Ag 15 nm, with insets displaying image of films deposited on glass substrates (e) Image of a semitransparent PSCs incorporating the BCP/Ag/MoO₃ electrode structure. Reproduced from Ref. [22] Copyright 2017 American Chemical Society

4. Challenges and Future Directions in B-PSCs

4.1 Challenges

4.1.1 Rear electrode transparency

Achieving a high degree of transparency in the rear electrode is essential for efficient bifacial light harvesting; however, it often comes at the expense of electrical conductivity. Materials like ITO and FTO offer good transparency but suffer from brittleness and limited scalability. Meanwhile, ultrathin metal films and transparent conductive polymers provide flexible alternatives but typically exhibit higher sheet resistance and reduced durability. Balancing optical transmittance, electrical performance, and long-term reliability remains a significant challenge.

4.1.2 Stability

Material stability is a critical concern in B-PSC. Although perovskite materials exhibit excellent optoelectronic properties, they are inherently vulnerable to environmental stressors. Moisture, oxygen, UV radiation, and thermal cycling can degrade the perovskite layer and adjacent interfaces,

leading to phase decomposition and ion migration. In bifacial configurations, where both sides are exposed to the environment, the risk of degradation is even more pronounced. This necessitates the development of robust material compositions and interface engineering strategies that enhance intrinsic stability without compromising device performance.

4.1.3 Encapsulation of dual-sided exposure

Conventional encapsulation techniques, designed primarily for monofacial cells, are inadequate for bifacial architectures. Bifacial devices require symmetrical, transparent encapsulation on both front and rear sides to allow unhindered light entry while ensuring mechanical protection and environmental sealing. Developing encapsulants that combine high optical clarity, low water vapor transmission rates, UV resistance, and mechanical flexibility is particularly challenging. Moreover, ensuring compatibility between the encapsulant and various device layers without introducing interfacial degradation pathways is a critical requirement.

4.1.4 Process scalability for large-area devices

Scaling B-PSC from laboratory-scale prototypes to industrially viable modules are hindered by challenges in maintaining uniformity over large areas. Achieving defect-free perovskite films with consistent thickness, grain size, and crystallinity is difficult at scale, particularly with solution-based deposition techniques. Additionally, large-area coating of transparent electrodes with low resistivity and high durability remains an unresolved issue. These fabrication inconsistencies lead to performance losses and reduced yield, limiting the commercial potential of bifacial perovskite technology.

4.2 Future Directions

4.2.1 Advanced light management structures

Optimizing light management is key to maximizing the bifacial gain in perovskite solar cells. Integrating nanophotonic structures, such as photonic crystals, diffraction gratings, or plasmonic nanoparticles, can manipulate light paths to enhance absorption in the active layer from both the front and rear sides. Additionally, the use of textured substrates or reflective coatings can increase albedo utilization, particularly in diffuse light environments. These advanced designs not only improve photocurrent generation but also contribute to angular insensitivity and energy yield enhancement under real-world conditions.

4.2.2 Development of durable transparent electrodes

Future research should focus on developing rear electrodes that simultaneously offer high transparency, low sheet resistance, and robust environmental stability. Emerging materials such as doped metal oxides (e.g., AZO, GZO), graphene composites, and silver nanowire networks are promising candidates. These alternatives must be optimized for compatibility with perovskite layers and scalable fabrication techniques. Multilayer hybrid electrodes that combine metallic and dielectric components may also provide a tunable platform to meet performance and durability criteria for bifacial operation.

4.2.3 Development of advanced perovskite materials and stabilization techniques

The future success of B-PSCs heavily depends on the development of perovskite materials with improved intrinsic stability and higher photoelectric conversion efficiency. Research is increasingly focused on compositional engineering, such as the incorporation of mixed cations (e.g., FA^+ , Cs^+) and halides (e.g., I^-/Br^-) to form more stable and defect-tolerant perovskite phases. Furthermore, the application of molecular cross-linking techniques has shown significant promise in enhancing the performance of solar cell [14,56,57]. Other promising approaches include surface passivation using 2D perovskite layers, incorporation of polymeric interlayers, and solvent engineering during film formation to achieve uniform crystal growth [58]. These combined strategies aim to produce high-efficiency, moisture- and UV-resistant films suitable for long-term outdoor bifacial operation.

4.2.4 Field testing and modeling under real conditions

To bridge the gap between lab-scale efficiency and practical performance, extensive field testing under varied climatic and albedo conditions is essential. Such studies should evaluate long-term stability, bifacial gain, and energy yield in diverse environments. Moreover, integrating real-time monitoring data with machine learning models can aid in predictive performance modeling, enabling the design of adaptive systems tailored to specific geographies [59]. These insights will guide material selection, module design, and system optimization for widespread deployment.

5. Conclusions

B-PSCs represent a promising avenue for next-generation photovoltaic technologies, offering the potential for higher energy yield through dual-sided light harvesting. Over the past decade, significant progress has been made in material design, device architecture, and optical management strategies, enabling notable gains in efficiency and bifacial performance. However, challenges such as environmental stability, scalable fabrication, and reliable encapsulation remain critical barriers to commercial deployment. Continued research into robust materials, advanced transparent electrodes, and real-world performance modeling will be essential to unlock the full potential of bifacial perovskite technology in practical energy applications.

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Conflict of Interest Statement

The author declares that there are no conflicts of interest related to the publication of this manuscript.

Author Contributions Statement

S.S. was responsible for the conceptualization, validation, formal analysis, investigation, resources, visualization, original draft preparation, review and editing of the manuscript, and project administration.

Data Availability Statement

No datasets were generated or analysed during the current study.

Ethics Statement

This study did not involve human participants, animals, or sensitive data, and therefore did not require ethical approval.

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